

SAC Steel Project Task 7.1.3

DEVELOPMENT OF FRACTURE TOUGHNESS REQUIREMENTS FOR WELD METALS IN SEISMIC APPLICATIONS

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ABSTRACT

Fracture of unreinforced welded moment frame connections subjected to simulated seismic loads, was caused by the initiation of fatigue cracks and their propagation to critical size. The fatigue cracks initiated at the web-to-flange intersection at the weld access hole, the valleys on the flame cut weld access hole surface, the weld toe and weld imperfections. Final fracture occurred when the fatigue crack extended unstably either in the base metal or in the weld metal. Final fracture is determined by the size of a crack, the stresses and strains acting on the crack and the fracture toughness of the material. This paper presents the methodology used to establish the necessary and sufficient fracture toughness requirement for weld metal used in seismic applications. The methodology was based on fracture mechanics principals and on empirical correlations. The proposed Charpy V-notch toughness is 40 ft-lb at 70°F and 20 ft-lb at 0°F for components subjected to +50°F and higher. This CVN requirement should preclude weld metal toughness from being a contributing factor to the fracture of unreinforced moment frame connections. Further improvements in the fracture performance of the connections must be accomplished by changes in design, detailing, fabrication and inspection.

INTRODUCTION

Full-scale welded moment frame connections have been tested under simulated seismic loads developed by the SAC Steel Project (Goel *et al.*, 1999; Fry *et al.*, 2000; Ricles *et al.*, 2000). These were conducted to determine if newly detailed unreinforced fully restrained connections can behave satisfactorily in future earthquakes. The base metal for beams and columns were produced to ASTM A572 Grade 50 specifications. The specimens were welded with either E70TG-K2 or E70T-6 electrodes. All tests were conducted at room temperature.

The simulated seismic loads induce high-strain low-cycle fatigue deformation in the welded joint. A failure analysis (Barsom and Pellegrino, 2000) of the specimens tested at the University of Michigan (Goel *et al.*, 1999) and at Lehigh University (Ricles *et al.*, 2000) demonstrated that fatigue cracks initiated and propagated in all the tested connections.

One of the many SAC Steel Project objectives was to develop fracture toughness requirements for weld metals to be used in welded moment frame connections. The following sections describe the methodology used to develop the SAC-recommended Charpy V-notch (CVN) toughness requirements for weld metals in seismic applications.

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DEVELOPMENT OF FRACTURE-TOUGHNESS REQUIREMENTS

The primary objective in the structural design of large complex structures such as bridges, ships, pressure vessels, aircraft and buildings, is to optimize the desired cost, performance and safety requirements. This objective is achieved by considering the relationships among design, material properties, fabrication, inspection, operation and maintenance, and the contribution of each of these factors to the performance of the structure. Several fracture-control guidelines minimize the possibility of fracture in structures: ¹proper design; ²the use of materials with adequate strength, ductility and fracture toughness; ³elimination or minimization of stress raisers; ⁴proper inspection; and, the like. When these general guidelines are integrated into specific requirements for a particular structure, they become part of a fracture-control plan. Therefore, a fracture control plan is a specific set of recommendations developed for a particular structure and should not be applied indiscriminately to other structures.

The magnitude and fluctuation of the applied stresses, the geometry of the structural details, constraint, fabrication and inspection affect material performance. For example, ductile materials may behave in a non-ductile manner when the structural details are highly constrained and / or contain severe stress raisers such as notches, cracks or fabrication defects. Fracture toughness is one of several properties that may affect the performances of the material and the structural connection. Fracture toughness of steels is a function of constraint, temperature and loading rate; high constraints, low temperatures and rapid loading rates decrease the fracture toughness value. Requiring high fracture toughness does not ensure adequate structural performance when the stresses and stress ranges are high, or the structural details are highly constrained or contain severe geometric stress raisers (e.g., notches, cracks or fabrication defects). The safety and reliability of cost-effective structures and / or structural components depend on the contribution of, and trade-off between, many factors including fracture toughness.

This paper presents the development of Charpy V-notch (CVN) fracture toughness requirements for weld metals in seismic applications. These requirements were developed in the absence of knowledge of these factors: ¹seismic demands (e.g., loads and deformations) for different building configurations and connection geometries; ²fracture-mechanics type fracture toughness (e.g., critical stress-intensity factors, K_{IC} , or crack-tip-opening displacements, δ_c) for the basemetals, heat-affected zones, or weld metals; and, ⁴fabrication and inspection requirements. At the time the fracture-toughness development was performed only 3 room-temperature δ_c values were available for 1 (E70TG-K2) of several filler metals. Consequently, the development of fracture-toughness requirements within the context of a fracture-control plan was not possible. The proposed fracture-toughness requirements for weld metals in seismic applications were therefore based on the following: 1) fundamental fatigue-crack-propagation and elastic-plastic fracture-toughness behaviors of steels and weld metals; 2) the δ_c value above which negligible fatigue life remains under full-reversal seismic deformations, thus the crack driving force, δ , should be kept below the δ_c value; and, 3) the δ_c value was converted to an equivalent CVN value.

The methodology used to develop the CVN fracture-toughness requirements for weld metals in seismic applications is detailed in the following sections. Future technical developments and an improved understanding of the factors that are integral parts of a fracture control plan for buildings subjected to seismic loads and deformations may modify, augment or replace the methodology and / or the proposed requirements.

FATIGUE CRACK PROPOGATION BEHAVIOR

Failure analysis of unreinforced welded moment frame connections subjected to simulated seismic loads during testing at the University of Michigan showed that fracture was caused by the initiation and propagation of fatigue cracks, as shown in Figure 1 (Barsom and Pellegrino, 2000).

The fatigue cracks initiated at the web-to-flange intersection at the weld access hole,¹ the valleys of the flame cut weld access hole surface,² the weld toe and³ weld imperfections.⁴ The applied cyclic loads increased the size of the fatigue crack until it reached a critical dimension where unstable crack extension severed the beam flange (Figure 2).

The fatigue cracks in all the tested specimens exhibited stable ductile tearing under the applied cyclic loads. Subsequent unstable crack extension was ductile in some specimens and brittle in others. Regardless of the mode of unstable crack extension, the critical crack size at fracture was large and the remaining fatigue life under the simulated seismic loads was negligible.

The fatigue-crack-propagation behavior for metals can be divided into 3 regions (Figure 3) (Barsom and Rolfe, 1999). The behavior in region I exhibits a "fatigue-threshold" cyclic stress-intensity-factor fluctuation, ΔK_{th} , below which cracks do not propagate under cyclic-stress fluctuations. Region II represents the fatigue-crack-propagation behavior above ΔK_{th} , which can be represented by Equation 1:

$$da / dN = A(\Delta K)^m \quad (1)$$

where a = crack length.

N = number of cycles.

ΔK = stress-intensity-factor fluctuation.

A and m are constants.

In region III the fatigue-crack growth per cycle is higher than predicted for region II. The data show that the rate of fatigue-crack growth increases and that under zero-to-tension loading (that is, $\Delta K = K_{max}$) this increase occurs at a constant value of crack-tip displacement, $\Delta \delta_T$, and at a corresponding stress-intensity-factor value ΔK_T , given by Equation 2:

$$\Delta \delta_T = (\Delta K_T)^2 / E \sigma_{ys} = 1.6 \times 10^{-3} \text{ in. (0.04mm)} \quad (2)$$

where ΔK_T = stress-intensity-factor-range value corresponding to onset of acceleration in fatigue-crack-growth rates.

E = Young's modulus.

σ_{ys} = yield strength (0.2 percent offset) (the available data indicate that the value of K_T can be predicted more closely by using a flow stress, σ_f , rather than σ_{ys} , where σ_f is the average of the yield and tensile strengths).

Acceleration of fatigue-crack-growth rates that determines the transition from region II to region III appears to be caused by the superposition of a ductile tear mechanism onto the mechanism of

cyclic subcritical crack extension, which leaves fatigue striations on the fracture surface. Ductile tear occurs when the strain at the tip of the crack reaches a critical value. Thus, the fatigue-rate transition from region II to region III depends on K_{max} and on the stress ratio, R . Most of the useful fatigue life is when the crack is in regions I and II. In region III cracks extend by large increments with each load cycle.

FATIGUE CRITICAL STRESS INTENSITY FACTOR

The first step in the development of fracture toughness for weld materials in seismic applications was to require the fracture toughness value to be higher than would be calculated from Equation (2). In other words, the fracture toughness must be high enough to ensure that fatigue crack extension under seismic loads would take full advantage of the behavior in region II, and would transition into the fast fatigue crack propagation region III without becoming an unstable fast running crack.

The yield strength, σ_{ys} , and tensile strength, σ_u , of the E70TG-K2 weld metal used to fabricate the University of Michigan moment frame connections were 76 and 90 ksi, respectively (Johnson, 2000). For a stress ratio, R , equal to zero, the minimum fatigue-critical stress-intensity-factor corresponding to the transition into region III, K_{CIII} , is given by Equation 3:

$$K_{CIII} = \star E \sigma_{ys} \delta_T \quad (3)$$

where $K_{CIII} = \Delta K_T = K_{max}$ for zero-to-tension loading, i.e., $R = 0$

E = Young's modulus = 29×10^6 psi

$\delta_T = 1.6 \times 10^{-3}$ in.

Therefore, $K_{CIII} \cong 60 \text{ ksi} \star \text{in.}$

This K_{CIII} value is conservative because it does not account for the elevation of the yield strength due to the triaxiality at the center one third length of the beam-to-column weld or the minor effect of compressive stresses on fatigue crack propagation under the fully reversed simulated seismic loads. Therefore, the true fatigue-critical stress-intensity-factor must be larger than 60 ksi \star in.

FRACTURE CRITICAL STRESS INTENSITY FACTOR

Examination of the fatigue cracks that initiated from weld imperfections in the moment frame connections tested at the University of Michigan (Barsom and Pellegrino, 2000) suggested that an estimate of the critical crack size for the weld metal was either a 0.5-inch deep part-through crack or about a 1.5-inch through-thickness crack. These crack sizes in combination with assumed effective stresses were used to estimate the fracture-critical stress-intensity-factor, K_c (i.e., fracture toughness) of the E70TG-K2 weld metal used to fabricate the University of Michigan connections.

Because the moment frame weldments in the University of Michigan tests were subjected to plastic deformation under the simulated seismic loads, the flow stress (Equation 4):

Fracture Toughness = K_{Ic} = fracture critical stress intensity factor

$$\sigma_{flow} = [\sigma_{yield} + \sigma_{tensile}] / 2 = 83 \text{ ksi} \quad (4)$$

was used to calculate the fracture critical stress intensity factor, K_{Ic} , from the relationship (Equation 5; Barsom and Rolfe, 1999):

$$K_{Ic} = 1.12 \sigma_{flow} \sqrt{\pi a_c} \quad (5)$$

For a part-through crack conservatively modeled as an edge crack (i.e., part-through crack having infinite surface length) with $a_c = 0.5$ in. and (Equation 6):

$$K_{Ic} = \sigma_{flow} \sqrt{\pi a_c} \quad (6)$$

For a through thickness crack with $2a_c = 1.5$ in. Thus, the estimated fracture critical stress intensity factor, K_{Ic} , values for E70TG-K2 weld metal are 117 and 127 ksi√in., respectively.

At the time this methodology to estimate fracture toughness requirements for weld metal in seismic applications was being developed, only 3 room temperature crack-tip-opening-displacement (CTOD) values were available. The 3 test specimens were from a single weldment made with E70TG-K2 filler metal. The weldment was part of the University of Michigan full-size specimen test program and was fabricated in an identical manner as the full-size moment frame connection specimens. The CTOD tests were conducted at the Edison Welding Institute (Johnson, 2000).

The 3 room temperature CTOD values for E70TG-K2 weld metal were 0.0019, 0.0043 and 0.0084 in. These values were converted to fracture critical stress-intensity factors, K_{Ic} , by using the relationship (Equation 7; Barsom and Rolfe, 1999):

$$K_{Ic} = \sqrt{1.7 \sigma_{flow} E \delta_c} \quad (7)$$

where $\sigma_{flow} = 83 \times 10^3$ psi

E = Young's modulus, psi

δ_c = critical crack tip opening displacement, CTOD, in.

[Note: Equations (2) and (7) are empirical correlations of K , σ and δ for fatigue and fracture, respectively (Barsom and Rolfe, 1999). Equation 2 can be changed to include the 1.7 constant in Equation (7). The value of $\delta_T = 1.6 \times 10^{-3}$ in., however, would have to be adjusted accordingly.]

The 3 CTOD values corresponded to K_{Ic} values of 88, 133 and 185 ksi√in. Consequently, assuming that one or more of the specimens tested at the University of Michigan contained weld metal having a $K_{Ic} = 88$ ksi√in., one may conclude that this fracture toughness value resulted in a large ductile fatigue crack prior to fracture (Barsom and Pellegrino, 2000).

The preceding discussion indicates that the fracture toughness, K_{Ic} , of E70TG-K2 weld metal from full-scale test specimens and from CTOD tests of weldment ranged from 88 to 185 ksi√in. Also, ductile crack propagation preceded unstable crack extension in the welded moment frame connections tested at the University of Michigan. Consequently, a minimum fracture toughness

requirement of 90 ksi√in. for weldments subjected to seismic loads was established. Higher fracture toughness values would have negligible beneficial contribution to the performance of the connections. Further improvements in the performance of welded moment frame connections must be achieved by improvements in connection design, detailing, fabrication and inspection.

DERIVATION OF EQUIVALENT CHARPY V-NOTCH IMPACT TOUGHNESS

The minimum K_{Ic} requirement of 90 ksi√in. was used to derive an equivalent impact Charpy V-notch (CVN) foot pound value that can be used as a screening test for weld metal. A correlation between CTOD data and impact CVN toughness does not exist. Therefore, a procedure was developed based on the general behavior of CTOD test results as a function of temperature and by evaluating existing K_{Ic} - CVN correlations.

CTOD values of structural steels increase as the test temperature increases. Initially the increase is gradual, and then accelerates rapidly within a test temperature zone where significant stable ductile tearing prior to unstable crack extension becomes visible with the naked eye on the fracture surface of the CTOD specimens (Figure 4; Barsom and Rolfe, 1999). This rapid increase in fracture toughness would have a minor beneficial effect on the fracture behavior of welded moment frame connections subjected to the severe demands of cyclic seismic loads.

Having defined a K_{Ic} of 90 ksi√in. to be the desired minimum fracture toughness, an equivalent CVN impact energy absorption value had to be established. An evaluation of existing correlations suggested that the Roberts-Newton correlation (Equation 8; Barsom and Rolfe, 1999) may be helpful. Extreme care should be exercised in the use of this correlation because it can produce erroneous results. This correlation is used here only because the K_{Ic} values calculated from the upper shelf impact CVN energy absorption appear to approximate the K_{Ic} value above which stable ductile (fibrous) tearing precedes unstable crack extension (Figure 5; Barsom and Rolfe, 1999).

$$K_{Ic} = 9.35(CVN, ft-lb)^{0.63} \quad (8)$$

Thus, a K_{Ic} equal to 90 ksi√in. would be equivalent to an impact CVN upper-shelf value of about 37 ft-lb. A conservative value of 40 ft-lb was selected. Based on experiences with other engineering structures, this impact CVN requirement appears to be conservative. If and when a better correlation is developed, the required CVN toughness value could be revisited.

PROPOSED CHARPY V-NOTCH REQUIREMENTS

All component tests conducted in the SAC Project have been conducted at room temperature. Thus, the results of these tests are applicable to interior framed buildings. The minimum interior operating temperature for buildings, as expressed by several participants in the SAC Steel Project, is +50°F. Considering the difference in loading rate between seismic and CVN impact loads and the temperature increase of weldments under seismic loads, CVN requirements at 70°F should be adequate for use at +50°F.

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The finite element analysis and strain measurements by Fry (2000) demonstrate that the strain demands on the weld material are very high even for the RBS specimens. These data show that the strain demand on the weld material is 8 times the yield strain for an unreinforced post-Northridge connection and is 5 times the yield strain for an RBS connection. Consequently, the CVN requirements should be equally applicable to both connections.

The significance of the present 20 ft-lb at -20°F requirement for a moment frame connection exposed to 50°F and higher is not obvious. Although no data are available to investigate the significance, the 40 ft-lb at 70°F requirement may be used to justify relaxing the low temperature requirement to at least 20 ft-lb at 0°F .

In summary, based on the discussions presented in the preceding section, it is proposed that the impact requirement for filler metals used in the fabrication of seismically loaded rigid moment frame connections be:

40 ft-lb at $+70^{\circ}\text{F}$

and

20 ft-lb at 0°F

For connections exposed to $+50^{\circ}\text{F}$ temperatures or higher. This CVN requirement should preclude weld-metal fracture toughness from being a contributing factor to the fracture of moment frame connections in seismic applications. Further improvements in the fracture performance of welded moment frame connections must be achieved by changes in design detailing, fabrication and inspection. Further research is needed to define the CVN requirements for connections exposed to temperatures below $+50^{\circ}\text{F}$.

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KEY WORDS

Seismic applications

Moment frames

Weld metal toughness

Development of toughness requirements

Charpy V-notch

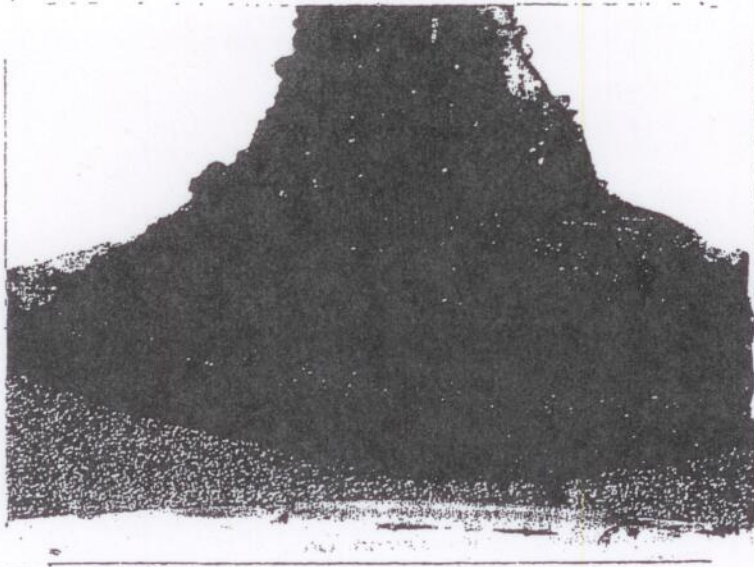


Figure 1. Initiation and Propagation of Fatigue Cracks in the University of Michigan Beam-to-column Test Welds.

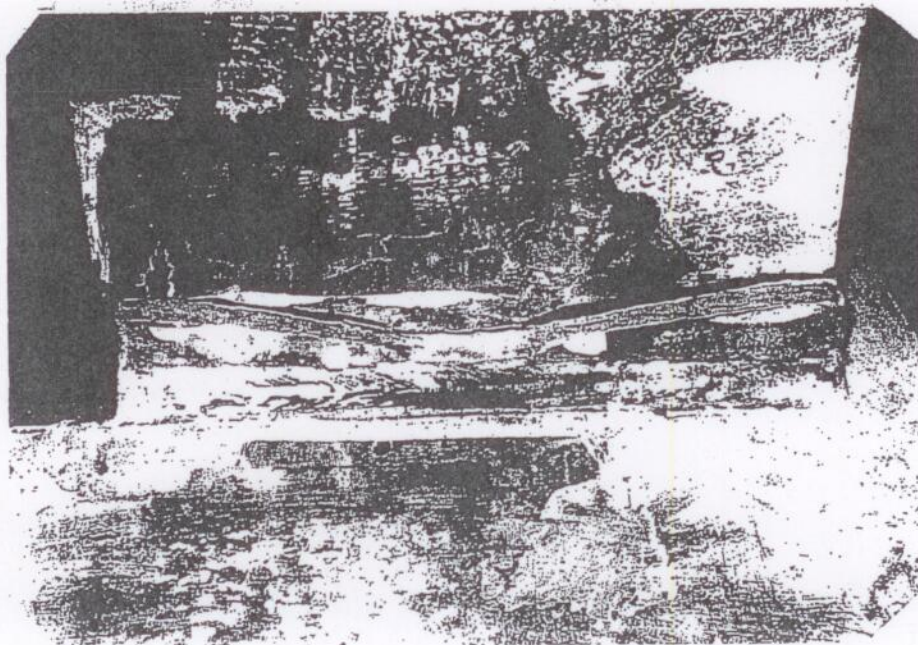


Figure 2. Stable Crack Growth (Center) Prior to Unstable Fracture.

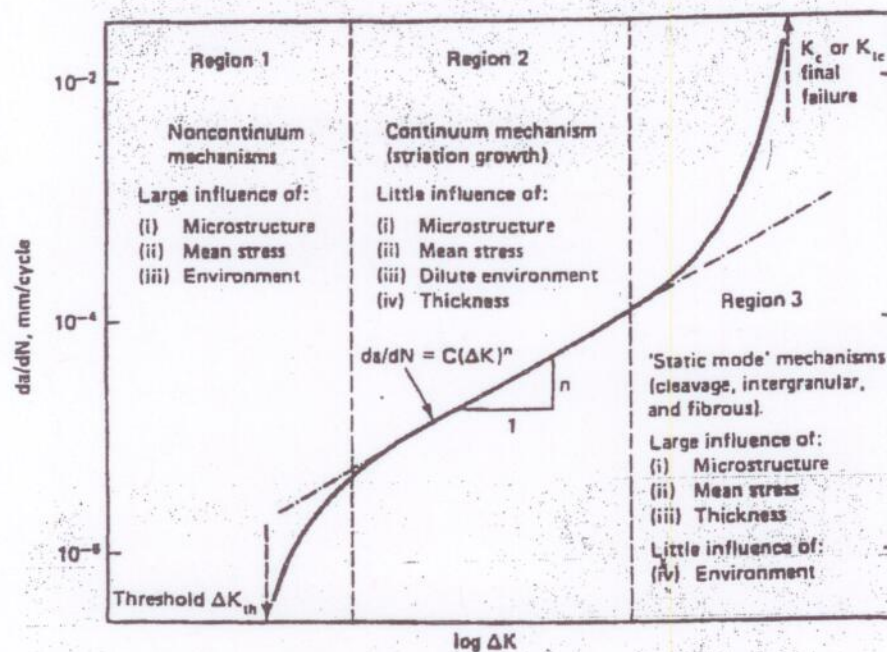
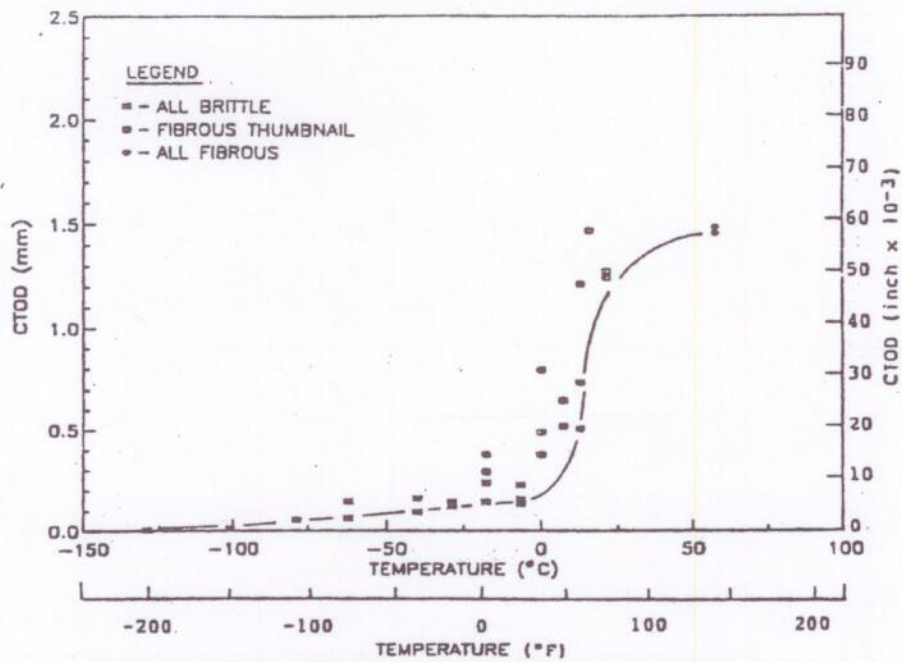
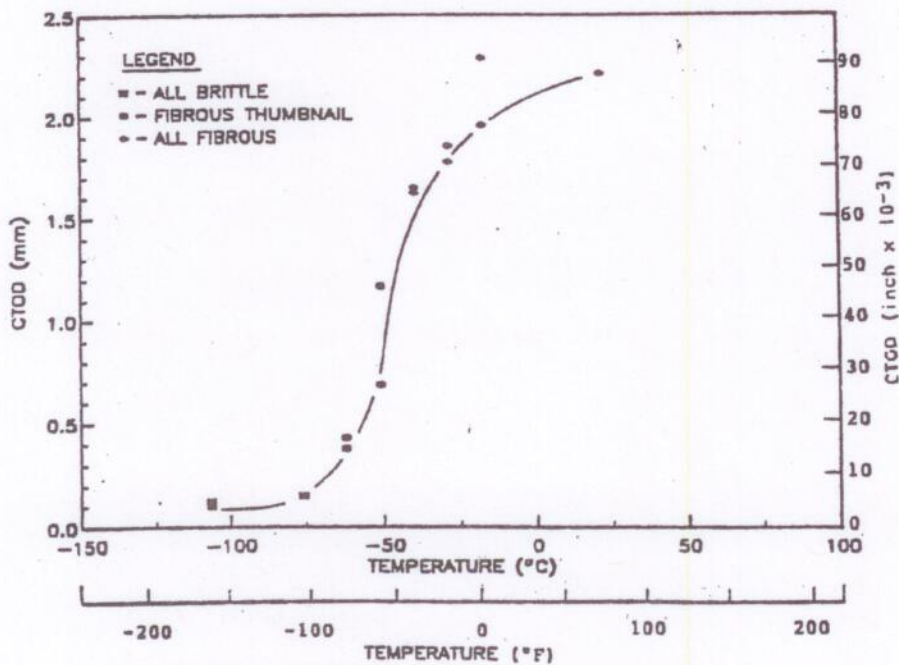


Figure 3. Schematic Representation of Fatigue Crack Growth Rate in Steels.

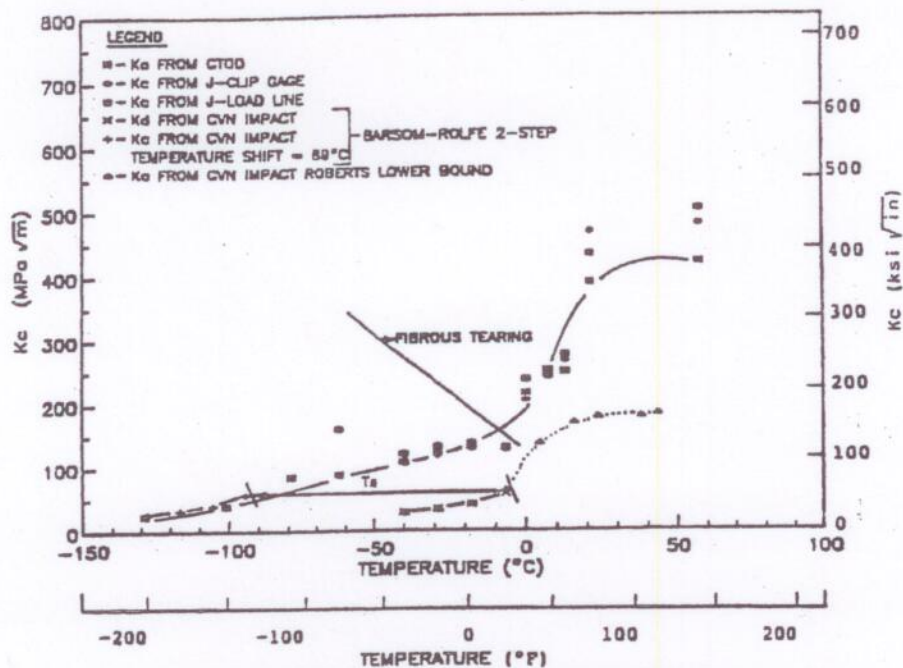


(a) CTOD-temperature transition curve for an A131 steel.

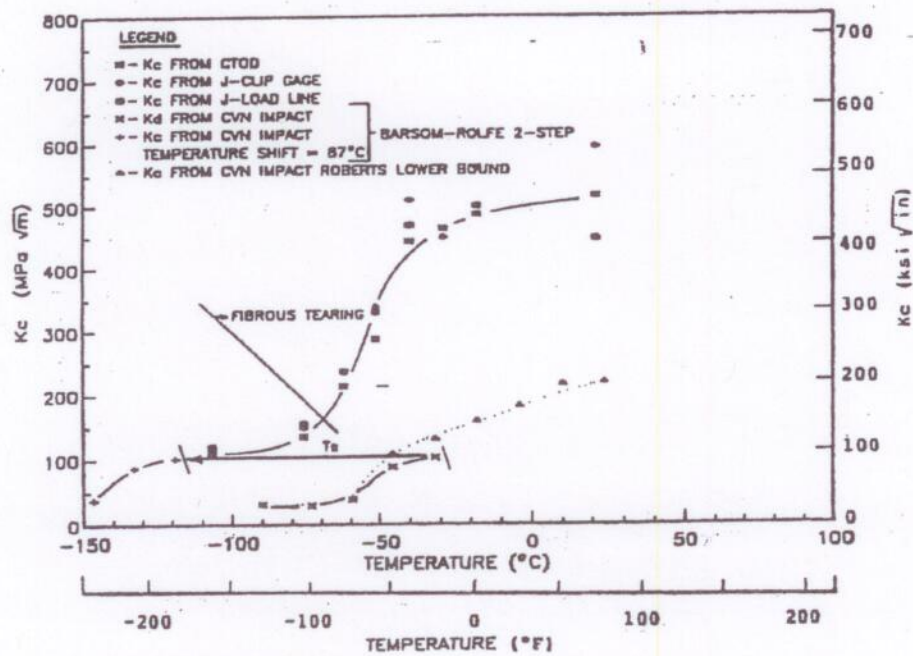


(b) CTOD-temperature transition curve for an A516 steel.

Figure 4. CTOD Temperature Transition Curves for Steels.



(a) K_{IC} -CVN-CTOD-J correlations for an A131 steel.



(b) K_{IC} -CVN-CTOD-J correlations for an A516 steel.

Figure 5. K_{IC} -CVN-CTOD-J correlations for steels.

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